

## Electrical conductivity and local structure of barium manganese iron vanadate glass

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**Abstract** Local structure and electrical conductivity of semiconducting  $20\text{BaO} \cdot 10\text{Fe}_2\text{O}_3 \cdot x\text{MnO}_2 \cdot (70-x)\text{V}_2\text{O}_5$  glass ( $x = 0-30$ ), abbreviated as  $x\text{BFMV}$ , were investigated by means of  $^{57}\text{Fe}$ -Mössbauer spectroscopy, differential thermal analysis (DTA) and DC four-probe method. Mössbauer spectrum of these vanadate glasses consists of a doublet with an identical isomer shift ( $\delta$ ) of  $0.38 \pm 0.01 \text{ mm s}^{-1}$ , indicating that distorted  $\text{FeO}_4$  tetrahedra constitute the structural units with distorted  $\text{VO}_4$  tetrahedra and  $\text{VO}_5$  pyramids. Quadrupole splitting ( $\Delta$ ) gradually increases from  $0.70 \pm 0.02$  to  $0.87 \pm 0.02 \text{ mm s}^{-1}$  with an increase in the  $\text{MnO}_2$  content, indicating an increased local distortion of  $\text{Fe}^{\text{III}}\text{O}_4$  tetrahedra. DTA study of these glasses showed a gradual increase of glass transition temperature ( $T_g$ ) from  $329 \pm 5$  to  $411 \pm 5^\circ\text{C}$ , showing an improved thermal durability. ' $T_g$  vs.  $\Delta$  plot' yielded a straight line with a large slope of  $707^\circ\text{C(K)}/\text{mm s}^{-1}$ , proving that  $\text{Fe}^{\text{III}}$  played a role of network former (NWF). An isothermal annealing of  $10\text{BFMV}$  glass at  $500^\circ\text{C}$  for 1000 min resulted in a marked increase in the electrical conductivity ( $\sigma$ ) from  $(4.5 \pm 3.9) \times 10^{-7}$  to  $(1.4 \pm 0.3) \times 10^{-2} \text{ S cm}^{-1}$  and a decrease in the activation energy for the electrical conduction ( $E_a$ ) from  $0.33 \pm 0.07$  to  $0.11 \pm 0.01 \text{ eV}$ , while  $\Delta$  of  $\text{Fe}^{\text{III}}$  decreased from  $0.76 \pm 0.02$  to  $0.49 \pm 0.02 \text{ mm s}^{-1}$ . These results suggest that decrease

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in the distortion of  $\text{Fe}^{\text{III}}\text{O}_4$  tetrahedra involved with the structural relaxation causes an increase in the probability of electron hopping from  $\text{V}^{\text{IV}}$  or  $\text{V}^{\text{III}}$  to  $\text{V}^{\text{V}}$ .

**Keywords** Barium manganese iron vanadate glass ·  $^{57}\text{Fe}$ -Mössbauer spectroscopy · Electrical conductivity · Structural relaxation

## 1 Introduction

Vanadate glass is known to be a semiconductor with an electrical conductivity ( $\sigma$ ) of  $10^{-7}$ – $10^{-5}$   $\text{S cm}^{-1}$ . The electrical conduction is known to be caused by polaron hopping from  $\text{V}^{\text{IV}}$  or  $\text{V}^{\text{III}}$  to  $\text{V}^{\text{V}}$  [1]. A drastic increase in  $\sigma$  was discovered in annealed barium iron vanadate glass [2] having a registered trademark of ‘NTA glass<sup>TM</sup>’ in Japan. Nishida reported a large charge-discharge capacity of 150 mAh  $\text{g}^{-1}$  in lithium-ion battery (LIB) in which cathode active material of annealed vanadate glass,  $\text{Li}_2\text{O}\text{-Fe}_2\text{O}_3\text{-V}_2\text{O}_5\text{-P}_2\text{O}_5$ , was used successfully [3]. This result indicates that vanadate glass could be a good candidate for the cathode active material of LIB. In the present study, substitution of manganese for vanadium was investigated in barium iron vanadate glass,  $\text{BaO}\text{-Fe}_2\text{O}_3\text{-V}_2\text{O}_5$ , in order to investigate the relationship between the local structure and the conductivity.

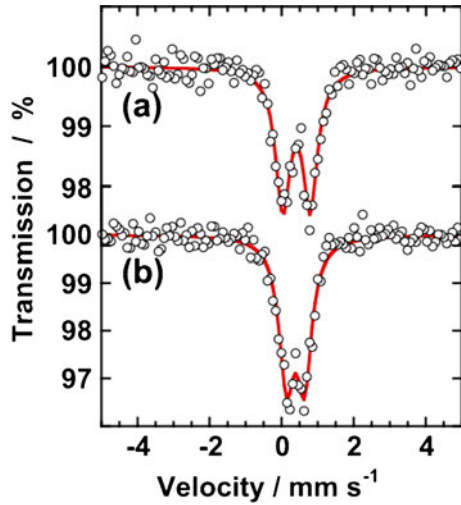
## 2 Experimental

Vanadate glasses with a composition of  $20\text{BaO}\cdot 10\text{Fe}_2\text{O}_3\cdot x\text{MnO}_2\cdot (70-x)\text{V}_2\text{O}_5$ , abbreviated as  $x\text{BFMV}$ , were prepared by a conventional melt-quenching method. Weighed amounts of  $\text{BaCO}_3$ ,  $\text{Fe}_2\text{O}_3$ ,  $\text{MnO}_2$  and  $\text{V}_2\text{O}_5$  of reagent grade were well mixed in a mortar and melted at  $1200^\circ\text{C}$  for 1 h in an electric muffle furnace. Homogeneous dark brown glass samples could be prepared when ‘ $x$ ’ was equal to or less than 30. Enriched isotope of  $^{57}\text{Fe}_2\text{O}_3$  ( $^{57}\text{Fe} = 95.54\%$ ) was used for some sample preparation. Isothermal annealing was carried out at  $500^\circ\text{C}$  for 1000 min. Mössbauer measurement was performed by a constant acceleration method with a source of  $^{57}\text{Co}(\text{Rh})$  and a reference of  $\alpha\text{-Fe}$  foil for isomer shift ( $\delta$ ). DTA was conducted from RT to  $500^\circ\text{C}$  under a heating rate of  $10^\circ\text{C min}^{-1}$  by using 10 mg of finely pulverized glass sample.  $\alpha\text{-Al}_2\text{O}_3$  was used as a reference of the temperature. Values of  $\sigma$  were measured at temperatures ranging from 30 to  $125^\circ\text{C}$  by dc-four probe method under the electric current from  $-1.0$  to  $1.0$  mA.

## 3 Results and discussion

Mössbauer spectra of  $10\text{BFMV}$  glass measured before and after isothermal annealing at  $500^\circ\text{C}$  for 1000 min, are shown in Fig. 1. When ‘ $x$ ’ was increased from 0 to 10, 20 and 30, consistent  $\delta$  value of  $0.38\pm 0.01$   $\text{mm s}^{-1}$  were observed, while quadrupole splitting ( $\Delta$ ) increased from 0.70 to 0.76, 0.79 and  $0.87\pm 0.02$   $\text{mm s}^{-1}$ . These results indicate that  $\text{Fe}^{\text{III}}$  atoms form distorted  $\text{Fe}^{\text{III}}\text{O}_4$  tetrahedra, and that they became more distorted when  $\text{MnO}_2$  was substituted for  $\text{V}_2\text{O}_5$ . In annealed samples, a marked decrease in  $\Delta$  was observed like 0.52 ( $x = 0$ ), 0.49 ( $x = 10$ ) and  $0.43\pm 0.02$   $\text{mm s}^{-1}$  ( $x = 20$ ), showing a largely decreased local distortion of  $\text{FeO}_4$

**Fig. 1** Mössbauer spectra of  $20\text{BaO}\cdot 10\text{Fe}_2\text{O}_3\cdot 10\text{MnO}_2\cdot 60\text{V}_2\text{O}_5$  glass of (a) before and (b) after isothermal annealing conducted at  $500^\circ\text{C}$  for 1000 min



tetrahedra involved in the structural relaxation. In case of annealed sample with  $x$  of 30,  $\Delta$  of  $0.54\pm 0.02\text{ mm s}^{-1}$  was observed, which is slightly larger than other annealed samples. This can be explained by the difference in the ionic radius of  $\text{Mn}^{\text{IV}}$  (39 pm) that is much smaller than that of  $\text{V}^{\text{IV}}$  (53 pm) [4]. It is considered that oxygen atoms constituting the network will be intensively attracted to  $\text{Mn}^{\text{IV}}$  in both as-quenched glass and annealed samples. Large  $\Delta$  of  $0.87\pm 0.02\text{ mm s}^{-1}$  obtained for as-quenched glass sample supports this idea. We can describe that chemical property of “vanadate glass” is predominantly observed when  $\text{MnO}_2$  content is equal to or less than 20 mol%.

DTA curves of  $x\text{BFMV}$  glass are depicted in Fig. 2. A gradual increase in glass transition temperature ( $T_g$ ) was observed from 329 to 347, 372 and  $411\pm 5^\circ\text{C}$ , while that in crystallization temperature ( $T_c$ ) from 378 to 400, 429 and  $465\pm 5^\circ\text{C}$ . A linear relationship found between  $T_g$  and  $\Delta$  of  $\text{Fe}^{\text{III}}$  was termed ‘ $T_g$ - $\Delta$  rule’ [5], i.e.:

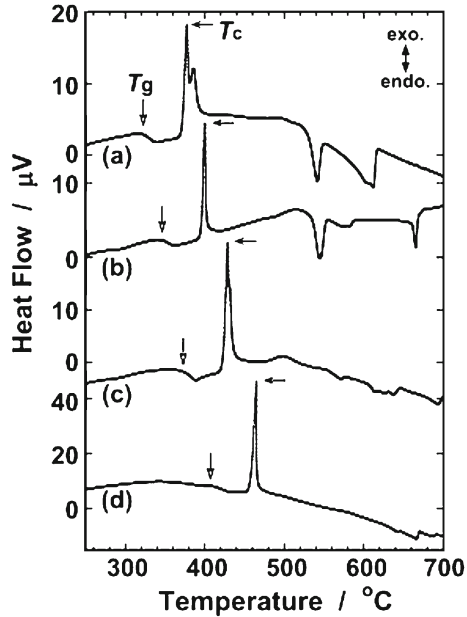
$$T_g = a\Delta + b \quad (1)$$

where  $a$  and  $b$  are slope and intercept of the straight line, respectively. According to the ‘ $T_g$ - $\Delta$  rule’ [5], large ‘ $a$ ’ value of  $680^\circ\text{C}(\text{K})/\text{mm s}^{-1}$  is generally obtained when  $\text{Fe}^{\text{III}}$  is located at tetrahedral site as network former (NWF), whereas ‘ $a$ ’ becomes only  $35^\circ\text{C}(\text{K})/\text{mm s}^{-1}$  when  $\text{Fe}^{\text{III}}$  is located at octahedral site as network modifier (NWM). In the present study, a large ‘ $a$ ’ of  $707^\circ\text{C}(\text{K})/\text{mm s}^{-1}$  was obtained, indicating that  $\text{Fe}^{\text{III}}$  atoms occupy tetrahedral site as NWF.

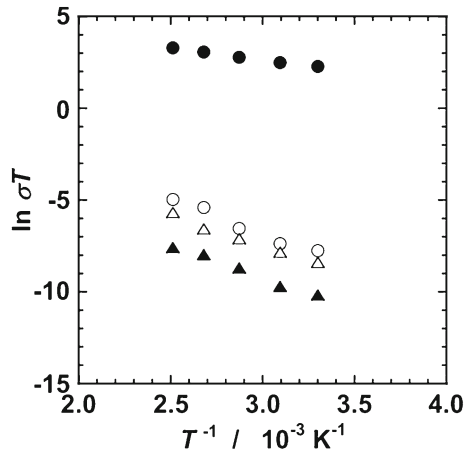
A slight decrease in  $\sigma$  value from  $(2.5\pm 1.2)\times 10^{-6}$  to  $(4.7\pm 3.8)\times 10^{-7}$ ,  $(4.2\pm 0.3)\times 10^{-7}$  and  $(2.3\pm 1.0)\times 10^{-7}\text{ S cm}^{-1}$  was observed when the ‘ $x$ ’ value of  $x\text{BFMV}$  glass increased from 0 to 10, 20, and 30. These results will be due to decrease in number of carriers ( $\text{V}^{\text{IV}}$  or  $\text{V}^{\text{III}}$ ) when  $\text{MnO}_2$  was substituted for  $\text{V}_2\text{O}_5$ . In the case of 10BFMV glass, a marked increase in  $\sigma$  was observed from  $(4.7\pm 3.8)\times 10^{-7}$  to  $(1.4\pm 0.3)\times 10^{-2}\text{ S cm}^{-1}$  after the annealing. According to the small polaron hopping theory [1], temperature dependence of  $\sigma$  is expressed by:

$$\sigma T = \sigma_0 \exp(-E_a/kT), \quad (2)$$

**Fig. 2** DTA curves of  $20\text{BaO}\cdot 10\text{Fe}_2\text{O}_3\cdot x\text{MnO}_2\cdot (70-x)\text{V}_2\text{O}_5$  glasses with 'x' of (a) 0, (b) 10, (c) 20 and (d) 30, recorded at a heating rate of  $10^\circ\text{C}/\text{min}$



**Fig. 3**  $\ln \sigma T$  vs.  $1/T$  plot of  $20\text{BaO}\cdot 10\text{Fe}_2\text{O}_3\cdot x\text{MnO}_2\cdot (70-x)\text{V}_2\text{O}_5$  glasses with x of 10 (circle) and 30 (triangle) measured before annealing (open symbols) and after isothermal annealing conducted at  $500^\circ\text{C}$  for 1000 min (closed symbols)



where  $E_a$  and  $k$  are activation energy for electric conduction and Boltzmann constant, respectively. As shown in Fig. 3, a drastic decrease in the  $E_a$  from  $0.33\pm 0.07$  to  $0.11\pm 0.01$  eV was observed after the annealing of 10BFMV glass. On the other hand, a constant  $E_a$  value of  $0.29\pm 0.06$  eV was obtained in the case of 30BFMV glass. It is interesting that the decrease in the local distortion of  $\text{Fe}^{\text{III}}\text{O}_4$  tetrahedra and a marked increase in the conductivity are clearly observed when 'x' is 10 or less, *i.e.*, the distortion of network involved with an increase in  $\sigma$  is intrinsic the vanadate glass.

## 4 Summary

Structure of  $x$ BFMV glass with  $x$  of 0, 10, 20 and 30 was investigated by means of  $^{57}\text{Fe}$ -Mössbauer and DTA.  $\text{Fe}^{\text{III}}$  atoms are incorporated in the glass matrix to form distorted  $\text{FeO}_4$  tetrahedra, and play a role of NWF at the substitutional sites of  $\text{VO}_4$  tetrahedra by sharing corner oxygen atoms with each other. Electrical conductivity of 10BFMV glass increased from  $4.7 \times 10^{-7}$  to  $1.4 \times 10^{-2}$   $\text{S cm}^{-1}$  after isothermal annealing conducted at  $500^\circ\text{C}$  for 1000 min. Increase in the conductivity is ascribed to a decrease in the distortion of  $\text{VO}_4$  and  $\text{FeO}_4$  units constituting the glass network, accompanied by a decrease in the activation energy for the electron hopping ( $E_a$ ) from 0.33 to 0.11 eV, and an increased probability of polaron (3d-electron) hopping from  $\text{V}^{\text{IV}}$  or  $\text{V}^{\text{III}}$  to  $\text{V}^{\text{IV}}$ .

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